

5                                    **COATED AND MAGNETIC PARTICLES AND APPLICATIONS THEREOF**

CROSS-REFERENCE TO RELATED APPLICATIONS

   This application claims the benefit of the filing of U.S. Provisional Patent Application Serial  
No. 60/431,315, entitled "Solid core solder particles for printable solder paste", filed on December 5,  
10                                    2002, U.S. Provisional Patent Application Serial No. 60/447,175, entitled "Electrochemical Devices and  
Processes", filed on February 12, 2003, and U.S. Provisional Patent Application Serial No. 60/519,813,  
entitled "Particle Coelectrodeposition", filed on November 12, 2003. This application is a continuation-in-  
part of U.S. Patent Application Serial No. 09/872,214, entitled "Submicron and Nanosize Particle  
Encapsulation by Electrochemical Process and Apparatus", filed May 31, 2001. The specifications of  
15                                    each application listed are incorporated herein by reference.

BACKGROUND OF THE INVENTION

Field of the Invention (Technical Field):

   The present invention relates to magnetic or nonmagnetic particles which are coated with a  
20                                    desired material before the particles are deposited on a substrate or surface. The coating is thus in  
intimate contact with the core particle, resulting in enhanced stoichiometric control and minimization of  
oxidation. The coating may be any desired material, including but not limited to a solder material or a  
catalytic material. If the core particles and/or the coating are magnetic, external magnetic fields may be  
used to enhance the deposition rate or to direct the particles to specific locations, minimizing deposition  
25                                    on unwanted areas of the substrate. Multiple types of deposition processes may be used, such as  
electrodeposition, screen printing, and photostencil bumping. The present invention also relates to the  
use of uncoated magnetic or nonmagnetic particles to modify the properties of other structures, such as  
semiconductor vias or bumps.

30                                    Background Art:

   Note that the following discussion refers to a number of publications by author(s) and year of  
publication, and that due to recent publication dates certain publications are not to be considered as prior  
art vis-a-vis the present invention. Discussion of such publications herein is given for more complete

background of the scientific principles and is not to be construed as an admission that such publications are prior art for patentability determination purposes.

5 Wafer bumping technology recently has attracted considerable attention in the high-end computing and networking markets, primarily because this technology has enabled high performance for high-density MPU, ASIC and memory device structures. Flip chip ball grid array (FCBGA) is a package type that uses solder bumping interconnection while simultaneously allowing for an area-array configuration. This ensures signal and power/ground integrity far superior to conventional peripheral wire bonding interconnection.

10 For commodity or consumer products, such as cellular phones, the package size is of vital importance. Chip scale packages (CSP) already are well accepted in the industry. However, the search is still on for even smaller solutions, such as wafer-level CSPs (WL-CSP), which are a true chip size package. Another example, bumped die for flip chip on board (FCOB) assembly, also can reduce a product's final size.

15 There are three primary wafer bumping processes: evaporation, electroplating and screen printing. Evaporation methods require substantial investment in capital equipment and typically entail high cost of ownership. Electroplating methods are known to drive the trend for finer bump pitch, but some solder materials are not suitable because of electroplating bath constraints. Screen printing methods typically are the most cost efficient; but there can be severe limitations on bump height when the bump pitch is less than 200 microns. Deposition processes that are useful according to the present invention also include electrodeposition, electrophoresis, photostencil bumping, and the like.

20 A fourth, recently developed advanced printing (photostencil bumping) bump method uses a photosensitive resist film and provides a solution that can address the entire range of applications, from consumer to very high end. This partly is due to the advantage of an advanced screen printing bumping process, which enables both a bump height comparable to electroplating methods and a cost structure that is competitive with standard screen printing. This method is ideal for wafer "shuttle" services, i.e., fabrication of different devices on a single wafer for one or many users who share the initial tooling costs. Because a "shuttle" wafer must be singulated prior to individual user or customer shipment, single chip solder bumping is an effective method to apply bumps separately on each device. Photostencil bumping has achieved a bump pitch as low as 100-micrometers and bumping on wafers as thin as 100 microns.

These advances allow for FCOB to become a very viable solution for system miniaturization. Naturally, an optimal solution also would have to consider the total cost for bumping, substrates, packaging, assembly, testing and board- level assembly. Photostencil bumping expands current capabilities for more uniform fine-pitch bumps. The height of a bump fabricated using photostencil bumping is similar to that enabled using electroplating and at a cost competitive to typical screen-printing methods. For example, photostencil bumping can produce bumps with a height of 105 ~ at a 200 micron pitch, while electroplating produces bumps 100 micron in height, and screen printing yields bumps only 75 micron tall. One important aspect of this technology is its use of a unique photosensitive resist film selected for its outstanding properties in patterning, as well as the fact that it can withstand the high temperatures required for bump formation while still responding well to stripping by alkaline solvents. Furthermore, due to the use of dry film openings at patterning, the height uniformity of the bumps is improved vastly.

The present invention may be used with any method of deposition, including all of the foregoing.

Key material, design and process considerations in solder bumping are as follows:

- 1) The bump material should ideally be high-temperature, eutectic-forming, and lead-free
- 2) The bump pitch should be as small as possible, taking into account substrate compatibility (Bismaleimide Triazine [BT], build-up, high-thermal expansion glass ceramic, etc.).
- 3) The bump height should be sufficient to ensure first-level reliability.
- 4) The bump configuration may be area-array (MPU/ASIC) or peripheral (memory/analog).
- 5) The bump process may be wafer-level (evaporation, electroplating, and screen printing) or single die (dimple plate).
- 6) Tailoring of melting point is desirable to accommodate multiple reflow processing steps.
- 7) Electrical contact testing must be performed both before and after assembly.
- 8) The bumps should have mechanical properties, such as strength, sufficient to withstand possible mechanical shock, vibration, creep, and fatigue occurring in some applications, thus ensuring long term reliability.
- 9) The material should be void-free before and after reflow.
- 10) Cost must be minimized.

The demand for lead-free bumping materials has increased because of the Waste Electric and Electronic Equipment (WEEE) & Restriction of Hazardous Substances (ROHS) Directive proposals in Europe. Furthermore, lead-free bumps minimize alpha-particle effects on memory macros in system-on-chip (SoC) devices. It has been demonstrated that as many as 11,000 bumps can be fabricated on a single die at 153 micron bump pitch using lead-free bumps on a copper wiring.

There is a need for new solder materials that have the characteristics described above. Elemental particles have been added to existing solder compositions in an attempt to improve these characteristics (see S. Jadhav et al., J. Electronic Materials 30 (9), 1197 (2002), F. Guo et al., J. Electronic Materials 30 (9) 1073 (2001), S. Hwang et al., J. Electronic Materials 31 (11) 1304 (2002), and S. Choi et al., J. Electronic Materials 28 (11) 1209 (1999), all of which are incorporated herein by reference). Uncoated particles have been electrodeposited along with a matrix or filler material. However, these approaches require multiple processing steps, increasing the complexity and cost.

In addition, solder pastes have been blended from elemental powders, but these have the disadvantages of poor shelf life, stratification in the paste (which greatly reduces uniformity and thus reliability), and the use of organic binders which are incompatible with some applications.

The present invention also relates to semiconductor fabrication techniques requiring the fill of blind vias with metallic features and the fabrication of termination devices in column or spherical shape require accelerated deposition of metals. The existing process utilizes electrodeposition, electroless deposition, plasma vapor deposition, and in some cases metallized screen printing inks and pastes.

A common technique is the electroplating fill of features defined by photoresist or photolithography. The electrodeposition occurs by metallizing the substrates and then under conventional electrodeposition steps the process of electrochemical deposition builds a metallic deposit in the defined feature until the amp minute requirement that controls the volume of fill is met. The real time process for these types of techniques varies from two hours to as much as ten or twelve hours to avoid occlusions or pinches in the feature that would sacrifice the full density structure.

The time involved in this process is not conducive to chemical or cost-effective processing. The features that result require a very complicated seed metallization to provide the current buss flow to carry out the electrodeposition. This process requires a complicated plasma vapor deposition of a seed metal

layer. This seed metal layer becomes very complicated to accomplish when the aspect ratio of the via feature exceeds 10 to 1. The current practice is to use more complicated methods of cross-sputtering and still the resulting result is not sufficient to assure a high-quality and cost-effective process.

5           The present invention also relates to the use of magnetic materials for catalysis. Membrane-electrode assembly (MEA) fabrication involves a great deal of often proprietary art, much of which has been developed by trial and error, to achieve the right combination of soluble Nafion, heat, and pressure for the proper interpenetration of PEM and catalyst that gives highly active catalyst layers. Typically inks of suspended precious metal blacks or carbon-supported precious metals are either brushed onto carbon  
10       felt electrodes or formed into catalyst decals by evaporation of catalyst inks on Teflon surfaces prior to pressure-transfer onto the PEM layer.

          The use of magnetic materials to enhance catalysis is known. The electrode fabrication approach taken by Leddy et al. relied upon blending carbon-supported and polymer-shrouded magnetic particles  
15       together with soluble Nafion to form ink, which results in considerable agglomeration of the magnetic particles and reduced contact with the separate electrocatalyst material surface (such as Pt). This method produces a catalyst layer containing a wide distribution of distances between magnetic and catalytic surfaces, yet test results were very attractive, demonstrating a three-fold improvement in the power levels compared to controls. Quantitative analysis of these results is complicated by lack of adequate  
20       knowledge of the microstructure of the magnetically modified catalyst layer, but it is estimated that only about 25% of the catalyst is active when CO is present.

          This approach has been limited to the use of relatively weak magnetic particles, because attempts to form catalyst layers from particles of higher magnetic strength failed to yield smooth,  
25       physically stable layers. High field-strength particles pose undesirable force affecting the electrodes integrity by fracturing and deforming bed-layer compact. Introduction of magnetic particles in the ink introduces new complications, such as how to apply and stabilize a thin layer while magnetic forces attract the particles together, during application and after drying of the applied ink. See for example Leddy, et al., U.S. Patent Nos. 5,817,221, 5,928,804, 6,001,248, 6,303,242, 6,322,676, and 6,479,176,  
30       the specifications of which are incorporated by reference. Leddy et al.'s method has other disadvantages, including segregation of each constituent resulting in non-uniformity of the final product and complex manufacturing process.

## SUMMARY OF THE INVENTION (DISCLOSURE OF THE INVENTION)

The present invention is both a method of depositing structures using coated and/or magnetic particles and the resulting structures. Deposition methods include but are not limited to electrodeposition, electrophoresis, electroplating, evaporation, screen printing, and photostencil bumping.

5

A primary advantage of the invention is that the stoichiometry of the resulting structure is more uniform than that of structures deposited using other methods due to the intimate contact of the coating and particle.

10

A primary advantage of the invention is that by depositing the particle and coating materials simultaneously, oxide contamination is minimized.

15

A primary object of the invention is to provide a method for making a solder joint by depositing coated particles on a substrate and reflowing the particles. The coating is preferably of a solderable material. The resulting solder may comprise unmelted particles in a solidified matrix. The presence of such particles reinforces the solder, making it more resistant to compressive and shear stresses. The particles also change the surface tension of a solder bump or similar structure, reducing the attainable bump pitch and enabling a higher density of bumps. The particles and coating may partially or completely react during reflow to form an alloy. The alloy preferably has a higher melting point than the coating, which permits subsequent multiple reflow steps.

20

25

A primary object of the invention is to deposit particles that are magnetic or that are coated with a magnetic material. The particles may be suspended in an ink or paste. Alternatively, the particles may be co-deposited in an electrolytic solution. The magnetic field is used to control the particle loading as well as precisely control the deposition location of the particles. In addition, materials with incompatible electropotentials may be deposited in one step.

30

A primary object of the invention is to permit the deposit of high aspect ratio structures, for example filling a via, without requiring complete seed metallization of the structure. A magnetic field may be employed to direct conducting particles into the structure past previously metallized surfaces, thereby forming an electrical contact and permitting the deposition to continue until completion.

5 The invention is further of a catalyst comprising a magnetic particle coated with a catalytic material. The presence of the magnetic field is known to improve the catalyst performance. The controlled geometry of a coating on a particle means the magnetic field at the surface is more easily controlled. The particle may optionally have at least one intermediate layer between the particle and outer coating, which acts as a diffusion barrier to prevent the magnetic particle from poisoning the catalyst.

10 Other objects, advantages and novel features, and further scope of applicability of the present invention will be set forth in part in the detailed description to follow, taken in conjunction with the accompanying drawings, and in part will become apparent to those skilled in the art upon examination of the following, or may be learned by practice of the invention. The objects and advantages of the invention may be realized and attained by means of the instrumentalities and combinations particularly pointed out in the appended claims.

#### 15 BRIEF DESCRIPTION OF THE DRAWINGS

The accompanying drawings, which are incorporated into and form a part of the specification, illustrate several embodiments of the present invention and, together with the description, serve to explain the principles of the invention. The drawings are only for the purpose of illustrating a preferred embodiment of the invention and are not to be construed as limiting the invention. In the drawings:

Figs. 1A-D depict two FCBGA package configurations before and after reflow;

25 Figs. 2A-B depict the effect of particle loading on bump shape;

Fig. 3A depicts PVD seed metallization of a via;

Figs. 3B-D depict partial PVD seed metallization and subsequent fill and etching of a via produced according to the present invention;

30 Fig. 4 is a schematic cross section of a coated catalytic powder which comprises an inner protective layer.

DESCRIPTION OF THE PREFERRED EMBODIMENTS  
(BEST MODES FOR CARRYING OUT THE INVENTION)

5           The present invention is of a solder material manufactured using coated particles. As used throughout the specification and claims, "solder material" also means filler metal, particle joining material, structural joining material, brazing material, welding material, and the like.

10           The coated powder preferably comprises a preferably metallic elemental core, for example nickel or copper, coated, optionally by electroplating, with a solderable metal or alloy such as tin, tin/lead tin/silver or other compositions suitable for electronic component joining. The core powder may be any size, including a few microns or even submicron, and thus can be made compatible with any manufacturing process. The core material may comprise nickel, copper or other conductive powder with a melting point higher than the melting point of the encapsulating deposit. The solder material may form  
15 a eutectic. The coating step may be accomplished without agglomeration. The coated powder may be electrodeposited onto the substrate as desired. Alternatively, the coated powder may be blended into a paste or ink that can be printed on to wafers through conventional photostencils, i.e., screen printed, and reflowed to create reinforced spherical bumps for, for example, flip chip bonding.

20           In this embodiment, during reflow the solder material melts, wetting and spreading over the entire core particle and substrate surface, and forming a solder bump. Upon solidification the solder coating joins together the individual particles and bonds with the substrate. The solidified bump contains embedded unmelted elemental core particles. Fig. 1 depicts two FCBGA package configurations: Fig. 1A depicts a bump for a glass ceramic substrate before reflow; Fig. 1B shows the same bump after  
25 reflow. Fig. 1C depicts a bump as deposited on a build-up substrate before reflow, and Fig. 1D shows the bump after etching and reflow. Figs. 1C and 1D show a composite solder bump comprising unmelted particles.

30           The resulting structure is a composite, or aggregate, material with a significantly higher compressive and shear strengths than solder that doesn't contain any particles. This reinforcement increases reliability not only because of resistance to mechanical shock, vibration, and the like, but also because the resulting bumps are strong enough to withstand shear stresses due to the mismatch in the



thermal expansion coefficients of the solder and the substrate, even at higher temperatures. This eliminates the need for underfill, saving cost and a manufacturing step.

Although the core material has not melted in this embodiment, there may be a reaction between the core particle and the coating. During reflow, a solid solution or intermetallic compound may be formed at the interface between the core particle and coating due to interdiffusion of the atoms in each. This diffusion may occur when the solder is liquid, resulting in a compositional change in the interfacial liquid, in which case the reaction is known as transient liquid phase bonding. Alternatively, if reflow is performed at a temperature less than the melting point of the solder coating, the reaction is called solid state diffusion bonding. In either case, the resulting interfacial compound may have a higher melting temperature than the original solder coating, thus enabling the material to withstand multiple reflow cycles at higher temperatures than the melting point of the original solder material, especially if all of the solder material has reacted with the core material. This bonding serves to drastically increase the strength of the solder joints. Traditional solder pastes composed of multiple elemental powders have the problem of non-uniformity due to the non-intimate contact of the reactants. These pastes thus cannot withstand multiple reflow cycles because they undergo secondary reflow due to prior incomplete reaction of their constituents. Multiple reflow capability is an important aspect of Level 1 soldering of an electronic device, because the device must be soldered further, joining on to the substrates of a mother board or electronic substrates as it is combined into a system.

In another embodiment, rather than using a core particle which doesn't melt during reflow, a powder material may be chosen which melts and thereby alloys with the coating during reflow. The alloy may be eutectic. For example, silver and copper particles have been coated with tin. The coated particles are screened onto a substrate. During reflow at a temperature higher than the temperature of both elemental metals, a eutectic alloy solder is created. Although these alloys are known, the advantage of this method is that they allow these materials to be deposited economically. Because for many alloys each of the elemental powders which form the alloy's components must be electrodeposited (that is, delivered ionically) in separate steps due to incompatibility of the electrolytic solutions, manufacturing time and cost is increased. In addition, there is a greater risk of oxidation of the surface of the already deposited powder because the substrate must be moved between multiple plating cells to deposit each element. Any oxide formation will inhibit the reaction of the elements. Another method entails evaporation of elemental multilayers which are then reacted; however, this process is very slow and expensive. These methods have the drawback of non-uniform stoichiometry of the final product due to

incomplete reaction and non-intimate contact of the reactant elements. Another prior method has been to deposit the material already alloyed, which requires very high reflow temperatures, greatly reducing solderability and possibly adversely impacting other device components which cannot withstand high temperatures. In all embodiments the coated powders of the present invention may be electrodeposited in a single step, thereby avoiding the aforementioned problems and reducing cost and manufacturing time. And because the coating and core powder are always in intimate contact, wetting is greatly enhanced and oxidation cannot occur.

The coated powders of the present invention can be used with any type of interconnections, for example Level 1 or chip-level interconnections, such as flip chip solder bumps, wire bonds, or stitch bonds, or Level 2 interconnections, which are the traditional printed circuit board solder joints including surface mount or through-hole configurations.

By matching the chemical and physical properties of the coatings and core powders, other desirable properties of the final material can be achieved, such as higher thermal conductivity to enhance package cooling and improved electrical properties such as current capacity or higher inductance if desired. And because adding particles changes the surface tension of the melted solder, at high particle loading (i.e. particle concentration or density), a lower bump pitch (i.e., increased bump density) can be achieved. Rather than forming a sphere during reflow, the bump may form an elliptical shape with steeper sides, which does not extend laterally to the same extent as a sphere, allowing bumps to be placed closer together. This is illustrated in Fig. 2, which schematically depicts the effect of particle loading on the reflowed solder bumps of Fig. 1. Fig. 2A is identical to Fig. 1D; as the particle loading is increased, the bump shape changes from a sphere to one having a narrower profile (Fig. 2B).

In addition, the final chemical composition of the material can be chosen to enhance the stability of properties such as electromigration of the solder material, for example pure tin. This will increase the reliability of the solder joint by, among other things, preventing the formation of solid state dendrites, which have been shown to cause gross failure in electronic components. These properties are applicable to any structure created according to the present invention, including but not limited to solder bumps, bond rings, and vias.

By choosing core particles which are magnetic, external magnetic fields can be utilized to enhance deposition of the solder, including but not limited to using the methods of electrodeposition,

photostencil bumping, and screen printing. The particles, or paste or ink that include them, can be more precisely directed exactly to the desired deposition location. The particle loading can be more precisely controlled. In addition, the unique properties of a magnetic core particle, made from an element or alloy such as nickel, could also have importance in novel fabrication techniques using magnetic field enhancements and the electrical testing of bumped die. For example, a magnetic field may be applied during reflow to control the spatial distribution of the powders in the solder joint, which may change the surface tension as well as wetting or other properties of the solder material.

The co-deposition of particles, preferably magnetic, can improve both the production process and final material properties of other microfabricated structures, such as vias, as well. Many of the advantages discussed above are applicable to these other structures. The particles may be uncoated particles, magnetic or nonmagnetic, or coated particles. The coated particles, if magnetic properties are desired, may have a magnetic core such as nickel. Alternatively they may comprise a nonmagnetic core coated with a magnetic material. By optionally using magnetic fields the particles can be more precisely directed to the desired deposition location. Also, dissimilar materials, for example those with widely differing electropotentials or incompatible electrolytic solutions, may be co-deposited in one step, saving time, manufacturing cost, and eliminating the potential for oxide contamination occurring between process steps. In addition, particle loading of the deposited material may be more precisely controlled using magnetic assistance. The final material may consist of the particles embedded in a matrix of the deposited material. The particles and/or the coatings thereon may react with the matrix material during further processing steps.

By co-depositing particles, preferably with magnetic assistance to more precisely direct the particles to the desired location, during fill electrodeposition, the current requirement for a complete film to be formed during prior seed layer metallization performed by PVD (plasma vapor deposition) could be relaxed. Thus the base of the via could be metal free and the particles being drawn in by the magnetic field would extend the electrical current buss into the base of the via, once they are in contact with a shoulder metallization that follows into the via. Fig. 3 shows the cross section of a via in silicon or ceramic substrate **300**. Fig. 3A shows PVD seed metallization **310** that was complete and coats the walls of the full three-dimensional geometry of the via. Fig. 3B depicts the same via geometry with PVD metallization **320** which is incomplete and tapers off towards the base of the via, leaving the base and possibly one-third to two-thirds of the length of the via without seed metallization.

Complete metallization would not be necessary, because by subsequently co-depositing conducting, preferably magnetic particles **330**, preferably by magnetic assistance, into the via during electrodeposition of the fill, they will extend the current flow into the base of the un-metallized via and provide the electrical continuity to provide a consistent reliable and repeatable electrodeposited fill of the via.

By introducing particles into the electrolyte and preferably directing them magnetically into the via, the rate of fill for the via can be accelerated linearly over a large range of particle concentrations; for example, a 60% solid concentration may increase the deposition rate by 60%, depending on the particle size and rate of loading. A typical volume ratio of particles to electrodeposit is approximately three to one, although other ratios are possible. As depicted in Fig. 3C, resulting fill **340** would be composed of particles **350** bound in the fully densified matrix of the electrodeposit. By co-depositing such particles, more favorable current conditions are created which allow acceleration of both the deposition and densification of the process.

According to Fig. 3D, wafer or substrate **360** may then be plasma etched on back side **370**, removing the substrate material and exposing bump **380** on the back surface of the wafer substrate which forms a through via interconnect. The rate of etch and the amount of substrate removed would define the geometry, including aspect ratio and the height, of the resulting bump on the back side of the wafer. In addition to the stated improvements in the via fill process, the presence of particles, preferably nickel particles, will also provide appreciable improvements in the thermal conductivity of the via and will provide a consistent solderable surface. Note that the drawings in Fig. 3 are schematic and are not meant to represent any particular relative size of the particles and via, or any particular particle concentration.

Another structure that may be deposited according to the present invention are bond rings, which are typically composed of a tin-gold eutectic solder. Preferably, 1-2 micron nickel particles are coated with tin, suspended in a gold electrolyte, and are co-deposited in a single step along with the gold. Multiple layers may be employed. Although a specific size range is disclosed, any particle size may be employed in order to optimize the properties of the structure. The magnitude and duration of an external magnetic field will partially determine the fill proportion and final composition of the deposited structure. After subsequent reaction a tin-gold composition may be formed, preferably an 80:20 eutectic composition. The nickel particles will mechanically reinforce the bond rings. Alternatively, pure tin particles suspended in the gold electrolyte may be co-deposited with the gold, again with the goal of

producing a desired eutectic composition. In the latter embodiment, magnetic fields would not be employed to assist with the co-deposition.

By selecting various particulate material, coated or uncoated, for its catalytic, electronic or other surface properties, the present invention can be used to create embedded passive component devices in a substrate during the microfabrication process. Such devices include but are not limited to resistors, capacitors and inductors. Choice of the particle, optional coating, and electrolyte materials would define further intrinsic properties that may be valuable in defining the properties of, for example, electronic components, hydrogen storage fields, and inductive or magnetic transducers.

#### Example — Magnetic core catalyst particles

An example of the use of coated magnetic core powders is in the manufacture of electrocatalyst materials for applications including, but not limited to, fuel cells. Not only does the use of magnetic core powders improve manufacturability of the device, it enhances its efficiency as well. The present invention comprises coating a magnetic particle, preferably Ni, with a catalytic material, preferably a metal, and preferably platinum. Optionally, other elements such as ruthenium may be added to the surface, either entirely encapsulating the particle or partially coating the surface, to tailor the catalyst's mechanical, electrochemical, electronic, and/or magnetic properties. The partial coatings may comprise isolated islands of the additional element. The ruthenium may optionally be oxidized.

The use of a magnetic material in catalyst electrodes results in improved catalytic properties. The magnetic moment of the core particles improves efficiency of device, and makes the catalyst more resistant to contamination. By fixing the electrocatalyst to the surface of the magnetic particle, and thus effectively providing a single distance from the magnetic material to the catalyst surface rather than a distribution of distances, a reasonable certainty to estimates of the local magnetic field at or through the catalytic surface is provided, so that a quantitative relationship can be established for the magnetic effect on CO tolerance. This is another advantage over the existing art.

One embodiment of the present invention is the production of magnetic electrode materials that can be cast on or pressed into ionomer membranes in a reliable and predictable fashion to give stable, uniform catalyst loading and membrane-electrode assemblies (MEAs) with highly active electrodes, even when utilizing high field-strength materials, such as Nd-Fe-B, are used that exert strong forces of self-attraction. The present invention provides superior MEA performance and tolerance to CO

levels present in hydrogen from reformed hydrocarbons, as well as improved abrasion resistance. An optional protective interfacial layer can render the core particles inert with respect with the catalytic reaction, and provide a robust interface ideal for addition of preferably Pt and/or Pt/Ru catalyst layers, directly on each encapsulated magnetic particle, where the field strength is the strongest. Preforming the precious metal layer onto magnetic particles completely encapsulated with a non-corroding metal bonding layer will place the electrocatalyst as close as possible to the surface of the magnetic material, regardless of subsequent processing steps to form MEAs. In the case of encapsulation by a protective Ni layer, or Ni-Pd layer if better corrosion resistance is needed, the protective barrier metal is also magnetic and should enhance the magnetic effect.

An example process for producing coated particles according to a preferred embodiment of the present invention is as follows.

1. Use optimal particle geometries from available metal and metal oxide powders that are suitable to process in aqueous electroplating solutions. Criteria should include magnetic saturation, geometry aspect ratio for dipole susceptibility, size/distribution, and surface morphology.
2. Determine a suitable barrier coat metal that can withstand the corrosive environment. Verify by acid test the required weight gain of the deposit to achieve full particle encapsulation without agglomeration.
3. Calculate the equivalent specific weight of platinum based on the previous catalyst loading reported. Determine parameters for electrodeposition of a very thin, uniform platinum coating on to the encapsulated magnetic particles, to give a 3 to 10 percent by weight platinum.
4. Optional deposition of a partial surface coverage of ruthenium onto the platinum coated encapsulated metallic particles, under conditions to yield a high degree of nucleation and formation of small islands of ruthenium on the surface of the particle.
5. Apply stable, uniform catalyst layers to membrane or gas-diffusion electrodes and form MEAs using Nafion 112 and demonstrate highly active electrodes layers.

6. Deposition of uniform layer of magnetically supported particles blended with Nafion polymers and mounted to carbon felt.

To ensure chemical and physical stability of the magnetic particles, we use inert metallic encapsulation techniques based on electrodeposition using a rotary-flow-thru electroplater that will provide magnetic beads encapsulated by a protective, corrosion-free barrier. The encapsulation process is based on electrodeposition using a patented rotary electroplater, specifically designed for electrolytic application of coatings onto particles in the few-micron to sub-micron range of diameters. One example of such a process is disclosed in U.S. Patent Application Serial No. 09/872,214, entitled "Submicron and Nanosize Particle Encapsulation by Electrochemical Process and Apparatus", filed May 31, 2001, incorporated herein by reference. This process is applicable to the manufacture of coated particles according to any of the embodiments of the present invention. A very durable platinum or palladium/nickel alloy coating may be applied and annealed to a nickel undercoat to keep the magnetic material from leaching into the cell and to put the Platinum or ruthenium catalytic element electrodeposited onto surface at the location of highest possible field strength. This approach provides a more robust, chemically inert layer than polystyrene, which is known to be unstable as the ionomer-base polymer in PEM fuel cells, PEM electrolyzers, or hydrocarbon reformers and will advance the development of the critical microstructure responsible for the beneficial effects of magnetic particles in MEAs. This method will provide an integrated composite of the materials and mitigate the uncertainty due to the art of blending electrocatalyst and magnetic materials.

The Rotary Flow-thru electrodeposition on powder encapsulation process utilizes centrifugal force to compact bulk materials in aqueous solution against an electrolytic cathode contact. The particle material is loaded through the top opening and the plating cell is rotated at sufficiently high rpm to centrifugally cast the powder against the cathode contact. Electroplating solution is continuously introduced at the top opening of the rotating cell through the immersed anode and flows through the cell exiting through a sintered porous plastic ring layered between the domed top, cathode contact ring, and base plate. Electroplating is carried out with a cycle of periodic stopping and/or counter rotation and sequential switching of the DC power supply to the cell to circulate the particle position for even coverage and prevention of agglomeration (bridging).

Optionally, the anode and cathode can be switched to operate the apparatus in anodic rather than cathodic mode. The sequential positioning of the nozzles, anodes (the anode can be easily

removed and switched to provide for deposition of different metals), and drain port provides a method to expose the materials being plated to a multiple step chemical process without intermixing the chemistry. Furthermore, the continuous immersion of the plated work prevents oxidation that normally occurs on the substrate when transferred from tank to tank in the conventional barrel plating process. The continuous immersion is preferably achieved by performing all steps of the process in the same cell. The chemical solutions are sequentially returned via the porous ring to the appropriate return drain for a discrete circulation of each chemical solution. Then by introducing the rinse water during high-speed rotation the chemical solutions are exchanged with minimal dilution due to the differing specific weights. Subsequent steps are then carried out.

The preferred cell process flow for electrolytic encapsulation of discrete particles with nickel plate (as an example) is as follows:

- 1 Load conductive powder;
- 2 Rinse;
- 3 Hot soak;
- 4 Nickel electroplate with start/stop cycle;
- 5 Rinse;
- 6 Hot rinse; and
- 7 Vacuum dry.

According to another embodiment, the Rotary Flow-Thru electrodeposition technique is used to encapsulate iron oxide (ferrite) powder to create a chemically inert magnet core, which will subsequently be rendered to inert permanent magnet beads with a platinum layer deposited on the nickel barrier. The process steps of this embodiment are as follows.

Use metal alloy powder with a particle size range of 3-5 $\mu$ m-diameters, which is electroplated in a nickel sulfamate solution at an amperage density of < 0.2 amps/dm<sup>2</sup>.

This material is then rinsed and dried in a vacuum oven for further processing.

The total amp hour requirement is controlled by weight gain percentage using the physical constants established to deposit nickel: 0.91308-ampere hours to deposit 1gram of 2-valency nickel metal.



After determining the weight gain percentage that assures chemical resistance and inertness. The platinum weight gain is determined by calculating specific weight of the active catalyst in an amount less than approximately  $0.4\text{mg}/\text{cm}^2$ . As a rule of thumb, the specific surface area goes down by about a factor of 3 for a 10-fold increase in particle radius, so somewhat lower Pt and Ru loadings may be required to keep the thickness of the catalyst layer to less than 20 microns for 1 to 5 micron diameter magnetic supports.

This applied equivalent weight is controlled by the physical constant for electrodepositing platinum metal: 0.54957-ampere hours to deposit 1gram of 4-valency platinum metal.

The resulting electroplated particles are examined using a scanning electron microscope and EPMA mapping of the electroplated platinum deposit, which measures Pt surface coverage, to verify complete and uniform Pt deposition. Should the Ni-encapsulated ferrite-based magnetic materials prove less stable than desired, either a different alloy may be used as the barrier layer, or higher loading of precious metal may be applied, or an alternative magnetic material, such as Ni-Fe or Al-Ni-Co, may be used.

After determining the coated particles meet the design specifications the particles can be permanently magnetized as powder with a medium energy (440 Joules), low voltage, capacitor discharge type magnetizer capable of saturating Alnico and Barium Ferrite magnetic materials.

A typical  $3\text{-}5\mu\text{m}$  diameter coated particle produced according to this embodiment is schematically depicted by the cross section in Figure 4. The particles are magnetically charged and ready for blending into ink for deployment as the catalyst electrode at either the cathode or anode. Note that the particles may be of any diameter, from submicron to over a hundred microns, depending on the requirements of the application.

The particle-size distribution, geometry, and degree of porosity may be determined by combined BET and Scanning electron Microscopy (SEM), coupled with energy dispersive X-ray analysis may be used to ascertain the depth of catalyst deposition and the purity of the applied platinum catalyst layer.

The quality of coatings is assessed by placing the particles in acid such as nitric acid, to determine if iron leaches from the core. If particles remain intact and no significant yellowing of the solution is observed, the particles are incorporated into a Nafion film on a glassy carbon electrode at ~15% loading.

5

The catalyst must now be applied to the electrodes. The direct application of catalyst layers to electrodes, which is the standard approach, is attractive from a commercial point of view, because of its compatibility with the demands of high volume manufacturing processes. Methods devised in the laboratory, such as decal transfer of ink layers cast first onto non-stick vellum, can be cost prohibitive in the real world. On the other hand, simple application methods, such as brushing of catalyst solutions directly onto porous gas-diffusion electrodes can be ill-defined and difficult to perform reproducibly to obtain the best degree of loading, penetration and uniformity. A simple, direct application method could yield spontaneous self-assembly of particles into the pores, driven by matching of relative particle sizes, when narrow size distributions are involved. Catalyst layers may be deposited as inks on membranes or carbon paper electrodes obtained from commercial sources as described in the literature, such as in a fashion according to the methods of Leddy et al., supra, to form a dense layer of catalyst particle in contact with the PEM layer. Tight tolerance on the thickness and uniformity of the catalyst layer may be achieved by controlling the viscosity of the ink solution, which is controlled by concentration of Nafion.

10

15

20

A slurry of catalyst layer may be prepared by dispersing the catalyst coated magnetic particles with carbon black Vulcan XC72R for the anode into the solvent substituted Nafion solution this was coated on the micro-porous layer formed electrode by tape casting. The particle distribution is assured by strong magnetic field on the backside of the casting surface.

25

One advantage of the present invention is the ability to use magnetic fields to assist in depositing a monolayer of particles on the MEA. This minimizes or eliminates the platinum which is not at the surface, and thus is not in direct contact with the flow stream, drastically reducing the cost of the device. In addition, magnetic fields may be used during screen printing to direct the deposition of catalytic particles, limiting them only to the pattern the flow stream will follow. Thus, the particles are not deposited where they will not be used, again dramatically reducing costs. For the present embodiment, the deposition method is preferably screen printing.

30

In addition to fuel cells, the present invention is applicable for batteries, including rechargeable batteries, hydrogen-based energy developers, electronics, and MEMS, delivering faster charge cycles, longer life, higher power, and smaller size.

5           A further embodiment of the present invention is to create multilayer or stratified compositions where on one layer a target material could be co-deposited followed by a second layer co-deposited with a reactive material, providing the capability to create solid state battery fields deposited on a substrate. By having the ability to deposit in stratified layers, many electrochemical devices, including fuel cell membranes, can be fabricated in the layered composition with the chemistry of the composition selected  
10           to perform the counter electrode properties of a normal electrochemical cell in this area.

          Although the invention has been described in detail with particular reference to these preferred embodiments, other embodiments can achieve the same results. Variations and modifications of the present invention will be obvious to those skilled in the art and it is intended to cover all such  
15           modifications and equivalents. The entire disclosures of all patents and publications cited above are hereby incorporated by reference.